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Two design strategies for enhancement of multilayer–DNA-origami folding: underwinding for specific intercalator rescue and staple-break positioning†

Yonggang Ke,^{‡,abc} Gaëtan Bellot,^{‡,abc} Niels V. Voigt,^d Elena Fradkov^{abc} and William M. Shih^{*abc}

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Single-layer DNA origami is an efficient method for programmable self-assembly of nanostructures approximating almost any desired two-dimensional shape from ~5 MDa of DNA building material. In this method, a 7 kilobase single “scaffold” strand is assembled with hundreds of oligodeoxyribonucleotide “staple” strands to form a parallel array of double helices. Multiple layers of such DNA sheets also can be designed to assemble into a stack, enabling the construction of solid three-dimensional shapes with considerably greater mechanical rigidity than two-dimensional shapes; however, the folding yield often is much lower and the required folding times are much longer. Here we introduce two strategies for designing multi-layer DNA origami that demonstrate potential for boosting assembly yield: (1) individual base pairs can be inserted between crossovers, allowing for greater bowing of helices at positions away from crossovers and therefore reduced electrostatic repulsion. At the same time, this underwinding of double helices increases a destabilizing torsional strain energy but then also increases affinity for intercalators, and binding of such intercalators can relieve this stress. We also have exploited this enhanced affinity for intercalators to PEGylate the surface of the nanostructures in a noncovalent fashion using PEG-tris-acridine. (2) Positioning of staple-strand breaks in the DNA origami such that each staple strand includes a 14 nucleotide (nt) continuous segment that binds to a complementary 14 nt continuous segment of the scaffold can greatly improve folding yields.

Introduction

Structural DNA nanotechnology, a method pioneered by Ned Seeman three decades ago, seeks the self-assembly of macromolecular shapes and devices while relying almost exclusively on Watson–Crick base pairing to program a specific association between component strands.^{1,2} The field has witnessed steady growth in the versatility and complexity of nanostructures that can be built. DNA origami in particular has served as a robust strategy for the construction of two- (2D) and three-dimensional (3D) shapes that are multiple megadaltons in size.^{3–6} Several applications are being explored for this rapidly developing technology, for example in molecular biophysics and structural

biology, biomimetic nanostructures (*e.g.* nanopore channels), photonics and energy, and therapeutics and diagnostics.⁷

Solid three-dimensional shapes can be achieved by folding stacked layers of DNA origami.⁴ However, yields of reported multi-layer designs have varied from a few percent to ~50% depending on the design, folding restrictions on monovalent *versus* divalent cation concentrations are much more stringent, and required folding times can be up to two orders of magnitude longer than what is necessary for single-layer DNA origami. This increased difficulty in folding multi-layer *versus* single-layer DNA origami may arise from the greater density of negative charge, access issues for components destined for the interior of solid 3D shapes, and more complicated strand crossover patterns due to the greater density of nearest-neighbor double helices. Many future applications will require reasonable yields of significantly more complex DNA nanostructures than currently are possible, which only will be attainable if optimization of the folding process can be achieved.⁷

In this study, we present two design strategies that show promise for the folding optimization of multi-layer DNA origami. In the first method, we design our nanostructures to have underwound double helices that are stabilized by the binding of intercalator dyes. This enhanced affinity of intercalators can be exploited for the design of functionalities that are

^aDepartment of Biological Chemistry and Molecular Pharmacology, Harvard Medical School, Boston, MA 02115, USA

^bDepartment of Cancer Biology, Dana-Farber Cancer Institute, Boston, MA 02115, USA. E-mail: William_Shih@dfci.harvard.edu

^cWyss Institute for Biologically Inspired Engineering at Harvard, Cambridge, MA 02138, USA

^dDanish National Research Foundation: Centre for DNA Nanotechnology at Interdisciplinary Nanoscience Center (iNANO), and Department of Chemistry, Aarhus University, DK-8000 Aarhus, Denmark

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‡ These authors contributed equally to this manuscript.

targeted to the underwound helices of the nanostructure *via* multivalent intercalation. In the second method, we explore the role of staple-break selection and find that avoidance of breaks in continuous 14 bp segments of the nanostructure can dramatically enhance folding. This result suggests that the corresponding 14 nt segments of the staple strands act as “seeds” that can nucleate proper folding of the desired target structure.

Results

Improved multi-layer DNA origami folding by specific intercalator rescue of underwound designs

Previously,⁵ we reported that a multi-layer DNA origami (60 helix bundle (hb) packed on a honeycomb lattice) folded at higher yields for a design where the initial reciprocal twist density of helices was programmed to be underwound at 11 bp turn⁻¹ *versus* the relaxed density of 10.5 bp turn⁻¹. The torsional stress was relieved somewhat by right-handed supertwisting of the bundle of double helices. The implication was that the destabilization from this underwinding was more than compensated by some other energetic gain. Underwinding was achieved by increasing the number of base pairs between crossovers (“targeted insertions”), thus the energetic gain may have derived from the increased ability of the negatively charged helices to bow away from each other, away from these crossovers, and thereby reduce electrostatic repulsion. Inserting base pairs in this way also increased the lengths of the staple strands, which may also help to improve folding.

Motivated by this previous result, here we sought to explore further the effects of “targeted insertions” on multi-layer DNA-origami folding yield. Towards this end, we generated a series of 24 hb multi-layer DNA origami where the helices are programmed *via* targeted insertions with initial reciprocal twist densities of 10.5 to 13.5 bp turn⁻¹ at 0.5 bp turn⁻¹ increments; therefore the total number of base pairs per design is proportional to the reciprocal twist density (*e.g.* 13.5 bp turn⁻¹ 24hb has 29% more base pairs than 10.5 bp turn⁻¹ 24hb) (Fig. 1A). In this design, the helices are arranged with their axes on a honeycomb lattice, in a manner that resembles a six-helix nanotube enclosed within an 18-helix nanotube (Fig. 1A). The average length of double helices for the 10.5 bp turn⁻¹ design is 77 bp, with a staple crossover density along each interhelical interface of one per 21 bp. The “head” of the bundle (Fig. 1A) is programmed to behave as a homodimerization motif (Fig. 1B, Figs. S1–S8†) for production of 24hb with double helices that have a length of 154 bp each (51 nm). Folding was achieved by mixing all component strands together, heating the mixture to 80 °C and cooling to 60 °C over the course of an hour, and then cooling to 24 °C over the course of 72 h. Divalent cations are critical for the folding of multi-layer DNA origami, therefore we folded each design at MgCl₂ concentrations of 6, 10, 14, and 18 mM; the free Mg²⁺ concentrations were 1 mM lower due to the presence of 1 mM EDTA added to chelate contaminating divalent cations. Folding was assayed by agarose-gel and transmission electron microscopy (TEM) analysis as the ability to form specific homodimers. We observed that the target homodimer exhibited a gel mobility intermediate between that for a misfolded monomer and a misfolded dimer (see below), as verified by gel

extraction and subsequent TEM imaging (data not shown). The discrepancy in gel mobility between the well-folded homodimer and the misfolded homodimer was fortuitous; otherwise, gel electrophoresis would have been much less informative.

Formation of the desired homodimer was most efficient for folding at 18 mM MgCl₂ concentration (Fig. 1D, first gel; TEM images not shown). Whereas the previously reported 60hb folded better at 11.0 bp turn⁻¹, here underwinding 24hb to 11.0 bp turn⁻¹ appeared to decrease the efficiency of folding, as indicated by the diffuseness and lower intensity of the product band (Fig. 1D, second gel; TEM images not shown). Underwinding 24hb to 11.5 bp turn⁻¹ gave rise to folding performance on par with 24hb underwound to 11.0 bp turn⁻¹ (Fig. 1D, third gel; TEM images not shown). However, underwinding 24hb to 12.0 bp turn⁻¹ abolished productive folding completely. Instead, a ladder of multimers of misfolded products was formed (gel results, Fig. 1D; TEM images not shown).

We hypothesized that for ≥ 12.0 bp turn⁻¹ underwinding, the penalty from torsional strain energy was greater than the energetic gain from increased helical bowing and decreased electrostatic repulsion. As positively charged intercalators can stabilize DNA in an underwound state, and furthermore can neutralize negative charge, we were motivated to investigate whether the presence of such intercalators in the folding mixture could rescue folding of the underwound 24hb designs (Fig. 1D). Therefore we refolded all the designs in the presence of 12 mM MgCl₂ and ethidium bromide at concentrations of 0, 8, 16, 32, 64, 128, 256, and 512 μ M (Fig. 1E). In order to saturate all potential binding sites for ethidium, assuming one site every two bp, a concentration of 10 nM particles, and ~ 1200 bp per particle, over 6 μ M of bound dye are required. However, if each intercalation event were to unwind a duplex by 26°, then the reciprocal twist density for such a hypothetically saturated dsDNA would be 17 bp turn⁻¹. The dissociation constant for ethidium binding to relaxed dsDNA is ~ 1 μ M,⁹ with stronger binding expected for underwound DNA but weaker binding expected for overwound DNA or DNA with greater occupancy of dye, due to charge neutralization and fewer available open sites.

Increasing concentrations of ethidium abrogated the ability of the 10.5 bp turn⁻¹ 24hb to fold, presumably due to preferential stabilization of DNA in an underwound state that only can be accommodated with a compensatory left-handed supertwisting that eventually becomes overly strained. In contrast, the 11.0 bp turn⁻¹ 24hb exhibited optimal folding at 8 μ M ethidium. This was consistent with our hypothesis that intercalator binding may improve folding by specific stabilization of the underwinding in the designed target structure relative to misfolded states, where underwinding is not enforced. At concentrations of ethidium higher than 8 μ M, folding of 11.0 bp turn⁻¹ 24hb may be inhibited by the same mechanism whereby ethidium counters the folding of 10.5 bp turn⁻¹ 24hb.

For 24hb underwound to 11.5 bp turn⁻¹ and 12.0 bp turn⁻¹, optimal folding occurred at 16 μ M ethidium. In this condition, the bands representing properly folded product are each significantly brighter than the corresponding band for the 10.5 bp turn⁻¹ design. We attribute this to two effects. First, the underwound designs have a greater number of ethidium molecules bound per particle, mainly due to greater affinity. Secondly, the 11.5 bp turn⁻¹ and 12.0 bp turn⁻¹ 24hb both

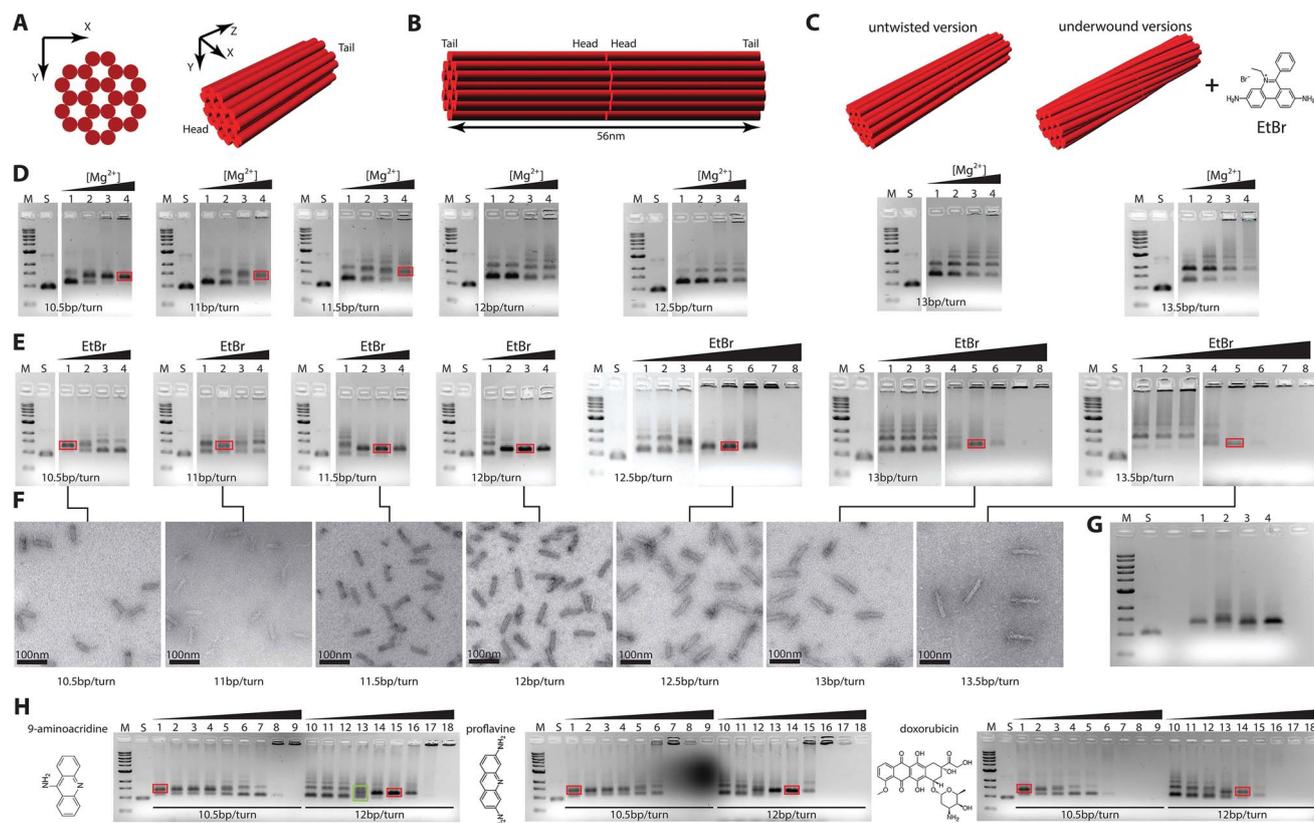


Fig. 1 Specific intercalator rescue of torsional-strain-induced misfolding of underwound DNA origami. (A) Design schematic of 24hb DNA origami folded with p3024 scaffold. Each cylinder represents a double helix. Left, view down helical axes. Right, rotated view. (B) Head-to-head dimer connected by connector staple strands. The yield of this dimer as indicated by agarose gel analysis was used as a measure of folding quality. (C) Gel electrophoresis assay of different versions of 24hb folded without intercalators. Lane M shows 1 kb DNA ladder (Fermentas). Lane S shows the p3024 scaffold. Lanes 1 to 4 show the 24hb samples annealed in the presence of 6, 10, 14, 18 mM MgCl_2 . The 24hb dimer bands circled in the red boxes correspond to best folded particles present. (D) Left, not underwound ($10.5 \text{ bp turn}^{-1}$) version; right, underwound version. The folding of low yield underwound versions can be improved by including DNA intercalators during the annealing process. (E) Gel electrophoresis assay of different versions of 24hb folded with EtBr and 10 mM MgCl_2 . Lane M shows 1 kb DNA ladder. Lane S shows the p3024 scaffold. Lanes 1 to 8 show 24hb samples annealed in the presence of 0, 8, 16, 32, 64, 128, 256, 512 μM EtBr. The 24hb dimer bands circled in the red boxes correspond to the best folded particles present in that sample. These circled materials were extracted and subjected to TEM analysis. (F) TEM images of well-folded 24hb dimers extracted from gels in (E). (G) Comparison of the folding of untwisted 24hb ($10.5 \text{ bp turn}^{-1}$) and underwound 24hb ($11, 11.5, 12 \text{ bp turn}^{-1}$) dimers. Each dimer was annealed under its optimal folding condition. Lane M shows 1 kb DNA ladder. Lane S shows the p3024 scaffold. Lanes 1 to 4 respectively show 24hb_10.5 bp turn^{-1} dimer folded with 18 mM MgCl_2 and no EtBr, 24hb_11 bp turn^{-1} dimer folded with 18 mM MgCl_2 and no EtBr, 24hb_11.5 bp turn^{-1} dimer folded with 10 mM MgCl_2 and 16 μM EtBr, and 24hb_12 bp turn^{-1} dimer with 10 mM MgCl_2 and 16 μM EtBr. (H) Folding of 24hb_10.5 bp turn^{-1} and 24hb_12 bp turn^{-1} dimers in the presence of other mono-intercalators. The 24hb dimer bands circled in the red boxes correspond to the best folded particles present in the sample. The green box indicates three bands that correspond to misfolded monomer, well folded dimer, and misfolded dimer, in order of decreasing mobility. Scale bars are 100 nm.

appear to fold to higher yield (estimated 24% for 24hb_11.5 bp turn^{-1} and 30% for 24hb_12 bp turn^{-1}) than the $10.5 \text{ bp turn}^{-1}$ 24hb (estimated 14%), as evidenced by the greater number of particles counted on the TEM grids (Fig. 1F), the lower amount of aggregated material in the gels of the analytical agarose gels under optimized MgCl_2 /ethidium concentrations, and the greater fraction of total lane intensity found in the major band of those gels (Fig. 1G, lane 1 *versus* lane 3 or 4). For 24hb underwound to $12.5 \text{ bp turn}^{-1}$, $13.0 \text{ bp turn}^{-1}$, or $13.5 \text{ bp turn}^{-1}$, optimal folding occurred at 64 μM ethidium bromide. However, at increasing concentrations of ethidium, aggregated folding products became more prevalent. Overall, the optimal yield was obtained with $12.0 \text{ bp turn}^{-1}$ 24hb folded at 10 mM MgCl_2 and 16 μM ethidium bromide.

Ethidium has been reported to underwind relaxed dsDNA by 26° per intercalation,⁸ therefore we can calculate the number of ethidium intercalations needed to stabilize the helices at the designed underwinding, shown as “Expected number of intercalations per helix” in Table 1. For example, one ethidium intercalation every six bp will yield a reciprocal twist density of 12 bp turn^{-1} in linear dsDNA, and one ethidium intercalation every 3.4 bp will yield a reciprocal twist density of $13.5 \text{ bp turn}^{-1}$ in linear dsDNA (the maximum binding occupancy for ethidium intercalation to dsDNA is one every 2 bp). We then measured the mean TEM length of each folded 24hb and used this to estimate the number of ethidium intercalators bound per helix, shown as “Apparent number of intercalation per helix” in Table 1.

Table 1 Expected *versus* measured EtBr intercalation events for 24hb DNA origami 10.5 bp turn⁻¹, 11 bp turn⁻¹, 11.5 bp turn⁻¹, 12 bp turn⁻¹, 12.5 bp turn⁻¹, 13 bp turn⁻¹, 13.5 bp turn⁻¹ versions

	24hb_10.5 bp turn ⁻¹	24hb_11 bp turn ⁻¹	24hb_11.5 bp turn ⁻¹	24hb_12 bp turn ⁻¹	24hb_12.5 bp turn ⁻¹	24hb_13 bp turn ⁻¹	24hb_13.5 bp turn ⁻¹
Base-pair per helix	154	161	169	176	183	191	198
Expected length without EtBr (nm)	52	54	57	59	62	65	67
Measured length for sample with EtBr (nm)	54	56	59	65	78	83	84
Expected number of ethidium intercalations per helix	0	9	20	29	38	49	58
Apparent observed number of ethidium intercalations per helix	6	6	6	18	47	53	50

We repeated these intercalator rescue experiments for 12 bp turn⁻¹ *versus* 10.5 bp turn⁻¹ 24hb using three other DNA-binding small molecules: 9-aminoacridine, proflavine, and doxorubicin. The results were qualitatively very similar to what was observed with ethidium, however the optimal concentrations were all higher (Fig. 1H). This is consistent with the lower reported affinities of proflavine and 9-aminoacridine,^{10–12} along with the lower amount of underwinding (17° per intercalation for 9-aminoacridine, proflavine, 10° per intercalation for doxorubicin) induced by the binding of these three relative to that for ethidium.¹³ A particular clear example of the mobility differences between the misfolded monomer, well-folded dimer, and misfolded dimer can be seen in the lane of the gel analyzing 12 bp turn⁻¹ 24hb folded in the presence of 9-aminoacridine (Fig. 1H, lane 13, lowest three bands outlined in green).

Functionalization of underwound DNA nanostructures with PEG-tris-acridine

The enhanced affinity of intercalators in underwound DNA nanostructures previously described can be exploited for the design of functionalities that are targeted to the underwound DNA helices *via* multivalent intercalation-based conjugation. Previous work has shown that multiple tethered intercalators exhibit avidity and form a more stable interaction with the double helix of DNA than a single intercalator.^{14,15} In addition, it has been described that tris-acridine has proper dimensions and good fitting for intercalation in the double helix of DNA.^{16,17} Therefore, we have synthesized a 20 kDa PEG molecule linked to tris-acridine in order to achieve better DNA binding affinity. The synthesis of the PEG-tris-acridine conjugates is described in Fig. S9† and involves two major steps: conjugation of 20 kDa PEG with dendrimer and reaction of dendrimer with acridine reagent. After each step, the isolation of products was achieved by size exclusion chromatography exploiting the large difference in molecular weight between reagents and by-products. Final purification was performed by reverse phase HPLC and good separation from non-reacted and other minor by-products was obtained. On the HPLC chromatogram, a stronger retention of PEG-tris-acridine was observed. Finally, mass spectrometry confirmed the synthesis of PEG-tris-acridine conjugates in the correct molecular weight range.

To test the DNA binding affinity of the PEG-tris-acridine, four distinct DNA nanostructures were designed using the honeycomb-lattice version of the caDNAno software:¹⁸ a six-

helix bundle at 10.5 bp turn⁻¹ (6hb_10.5 bp turn⁻¹), a twelve-helix bundle at 10.5 bp turn⁻¹ (12hb_10.5 bp turn⁻¹), a 12hb_11 bp turn⁻¹, and a 6 hb ring. The designs are illustrated and detailed in Fig. 2A, B and Figs. S10–S12†. Optimal MgCl₂ concentrations for folding were determined, and then used for assembling each DNA origami object in a one-pot reaction. Folding of the four nanostructures resulted in products that migrate as sharp bands on a 2% Mg-agarose gel. To further verify structural integrity, DNA-origami nanostructures were extracted from the gels and visualized by TEM. The TEM micrographs in Fig. 2B indicate successful formation of mono-disperse and homogeneous DNA nanostructures.

Next, we investigated the binding capabilities of the synthesized PEG-tris-acridine on single-stranded DNA (ssDNA) and DNA origami. The intercalation complexes were prepared by pre-mixing DNA with synthesized HPLC-purified PEG-tris-acridine at different ratios and incubated at 25 °C for 30 min. The binding efficiencies of PEG-tris-acridine conjugates with the DNA structures were analyzed by gel-mobility-shift assays using Mg-agarose electrophoresis. Incubated objects were electrophoresed for 3 h on a 2% agarose gel containing 0.5× TBE buffer and 11 mM MgCl₂. The samples were run on an ethidium-bromide-free gel in order to prevent intercalation competition between the ethidium and the PEG-tris-acridine. After electrophoresis, the gel was stained by ethidium bromide and subjected to image analysis on a Typhoon biomolecular imager following the fluorescence of ethidium bromide and acridine on a gel-shift assay (Fig. 2B).

First, we tested the binding efficiency of PEG-tris-acridine to ssDNA. M13mp18 viral DNA (8064 nt), which often is used as a scaffold strand in the assembly of DNA origami, was selected for this assay. The DNA scaffold (8064 nt) at 2 nM was incubated with different amounts of HPLC-purified PEG-tris-acridine conjugates in 40 μL reaction for 30 min at room temperature. The synthesized PEG-tris-acridine retarded the agarose-gel mobility of the ssDNA, as evidenced by the disappearance of the representative sharp DNA band (Fig. 2B). As shown in the gel in Fig. 2B, the intercalation of the PEG-tris-acridine molecule drastically reduced the DNA gel mobility at a weight ratio of PEG-tris-acridine to DNA of 5 : 1 or higher. By contrast, equivalent amounts of individual PEG(20 kDa) or tris-acridine demonstrated no influence on ssDNA mobility (data not shown). These results suggest that only the binding of PEG-tris-acridine conjugate with DNA resulted in band retardation during electrophoresis.

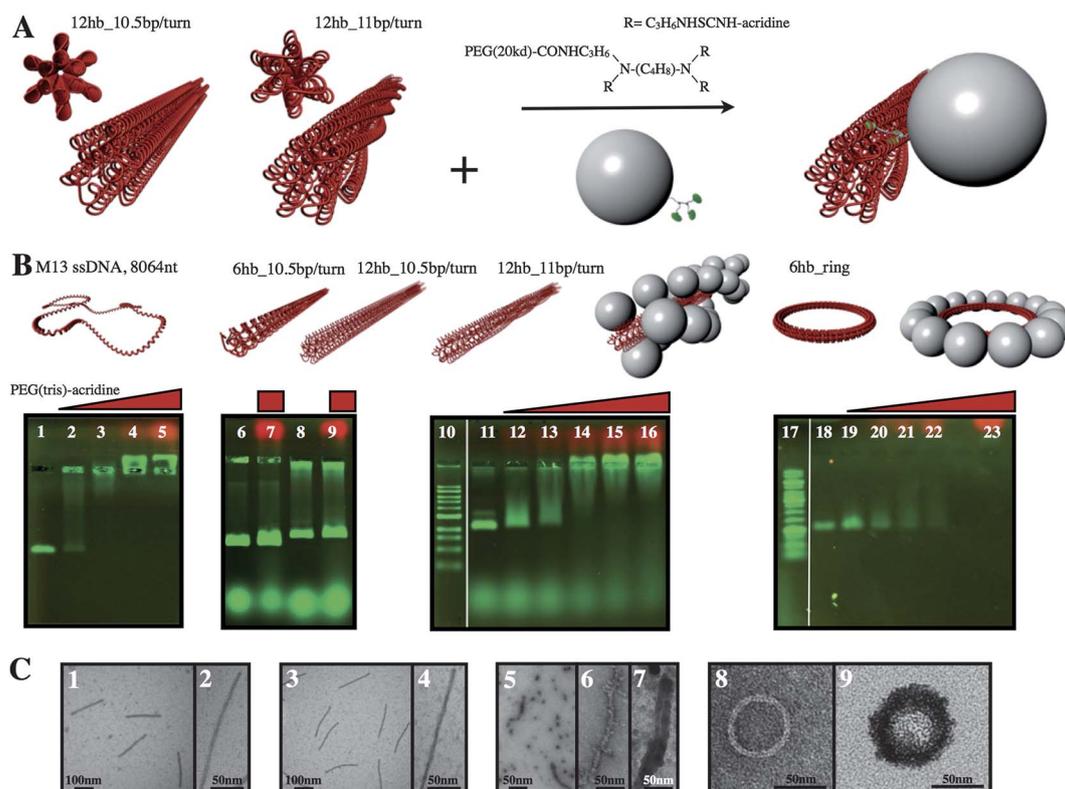


Fig. 2 Specific PEG-tris-acridine functionalization by multivalent intercalation into underwound DNA origami. (A) Design schematic of PEG-tris-acridine targeting the underwound helices of the twelve-helix DNA bundle. Left, model views off perpendicular and along to the helical axis of 200 nm-long twelve-helix DNA bundles with 10.5 and 11 bp turn⁻¹ reciprocal double-helical twist density. Right, model of the multivalent intercalation between underwound twelve-helix DNA bundle and synthesized PEG-tris-acridine. (B) Gel-mobility-shift assays using Mg-agarose electrophoresis following the DNA binding with PEG-tris-acridine at different weight ratios. The gels were stained by ethidium bromide after 1% (w/v) agarose gel electrophoresis. In green, the ethidium fluorescence from the DNA and in red, the acridine fluorescence from the PEG-tris-acridine. Lanes 1 to 5 show the retention of the p8064 M13-derived ssDNA at various PEG-tris-acridine to DNA weight ratios of 0 : 1, 2 : 1, 5 : 1, 10 : 1, 15 : 1. Lane 6, 6hb_10.5 bp turn⁻¹ without PEG-tris-acridine. Lane 7, 6hb_10.5 bp turn⁻¹ with PEG-tris-acridine at 60 μM. Lane 8, 12hb_10.5 bp turn⁻¹ without PEG-tris-acridine. Lane 9, 12hb_10.5 bp turn⁻¹ with PEG-tris-acridine at 60 μM. Lane 10, 1 kb ladder. Lane 11, underwound 12hb_11 bp turn⁻¹ without PEG-tris-acridine. Lanes 12 to 16 show the retention of underwound 12hb_11 bp turn⁻¹ at various PEG-tris-acridine to DNA weight ratio of 3 : 2, 4 : 1, 6 : 1, 8 : 1, 12 : 1. Lane 17, 1 kb ladder. Lanes 18, 6hb_ring. Lanes 19 to 23 show the retention of 6hb_ring at various PEG-tris-acridine to DNA weight ratios from 3 : 2, 4 : 1, 6 : 1, 8 : 1, 12 : 1. (C) TEM images of unPEGylated and PEGylated DNA nanostructures. 1, 12hb_10.5 bp turn⁻¹ extracted from gels and 2, zoom in images. 3, 12hb_11 bp turn⁻¹ extracted from gels and 4, zoom in images. 5, PEGylated 12hb_11 bp turn⁻¹, 6, zoom in images and 7, stained with 0.5% ruthenium tetroxide. 8, 6hb_ring unPEGylated. 9, 6hb_ring PEGylated. Scale bars are 50 nm.

We next explored the ability of the PEG-tris-acridine to bind folded DNA-origami nanostructures. To prevent further PEG-tris-acridine binding to ssDNA portions of DNA origami, each DNA-origami nanostructure was designed with capping strands (Figs. S10–S12†). The 6hb_10.5 bp turn⁻¹ and 12hb_10.5 bp turn⁻¹ nanostructures were folded with 10 nM scaffold and then were incubated with different amounts of HPLC-purified PEG-tris-acridine conjugates in a 40 μL reaction volume for 30 min at room temperature. After electrophoresis, no DNA gel-shift was observed even for mixtures with PEG-tris-acridine at 60 μM (*i.e.* 6000 times molar excess) (Fig. 2B). Even after incubation for 24 h and higher incubation temperature (50 °C), no evidence of binding was observed.

To assess whether global twisting could improve PEG-tris-acridine binding on DNA-origami, we designed a new version of 12hb in which we added a single base pair to every 21 bp, resulting in a shape with double helices underwound to an

average initial reciprocal twist density of 11 bp turn⁻¹; such structures relax some of the resulting torsional strain by right-handed supercoiling, similar to the above designs with 24hb (Fig. 2A, B). For this 12hb_11 bp turn⁻¹ design, significant gel retardation was observed for samples incubated in the presence of PEG-tris-acridine conjugates at a weight ratio of PEG-tris-acridine:DNA of 6 : 1 or greater, with partial retardation already observed at a ratio of 1.5 : 1 (Fig. 2B).

We next designed a 6hb_ring structure, with a 46 nm inner diameter and a 60 nm outer diameter, by implementing a balanced gradient of insertions in the outermost double helices and deletions on the innermost double helices to produce global bend with no global twist.³ After the addition of PEG-tris-acridine at different ratios to 6hb_ring, the mixture was incubated at room temperature for 30 min and analyzed by agarose gel electrophoresis as before. A similar gel retardation was observed to that seen for underwound 12hb at PEG-tris-acridine : DNA mass ratios of 8 : 1 or greater.

To provide direct visualization of the PEG-tris-acridine binding, the DNA nanostructures were characterized by negative-stain transmission electron microscopy (TEM) (Fig. 2C). We first visualized the DNA/PEG complex by using uranyl formate staining. For both successfully PEGylated nanostructures (12hb_11bp turn⁻¹ and 6hb_ring), we were able to detect DNA nanoparticles displaying surface features suggesting PEGylation, however the staining of these features was not robust or easily reproducible. More robust staining of the PEG functionalities could be seen with 0.5% ruthenium tetroxide. Ruthenium tetroxide is an aggressive oxidant, and exposure of the sample to the 80 keV electron beam often lead to visible damage on the carbon grid making the data collection more difficult. However, several areas on the carbon grid can appear preserved and intact. In Fig. 2B, it is clearly possible to see in both PEGylated nanostructures examples of specific PEG staining surrounding the DNA origami.

For 12hb_11bp turn⁻¹, a change in the diameter of the DNA rod was evident that was not seen in the absence of treatment with PEG-tris-acridine. The formula relating the molecular weight (M_w) of PEG to its radius of gyration (R_g) is $R_g = 0.215M_w^{0.583}$ Å, which is 70 Å for a 20 kd PEG polymer.¹⁹ PEG polymer can be represented as a sphere of radius R_g attached to the surface of the DNA-origami walls (Fig. 2A). The thickness of 12hb coated with PEG-tris-acridine appeared to be roughly consistent with the 35 nm predicted by this model; we can expect that the globular shape of the PEG polymer is deformed to some extent both by interactions with the DNA surface as well as due to staining and drying artifacts.

For the 6hb_ring, the PEG polymer appeared to localize to the outer double helices, where base pairs were inserted to exert bending of the bundle. Here the DNA is predicted to be locally underwound, thus it is as expected to find the PEG-tris-acridine binding favorably only on this surface. According to the TEM images, a global diameter of 84 nm of the 6hb_ring was observed, compared to 60 nm for the unPEGylated 6hb_ring, corresponding to an effective diameter of 12 nm for the PEG molecule, which is coherent with the diameter predicted.

Staple-break positioning for improved folding

As a second strategy for improving folding yield of multi-layer DNA origami, we investigated the effects of altering the staple-break distribution pattern in 24hb folded with 308 bp (336 nt accounting for the ssDNA ends) long double helices and the p8064 scaffold (Fig. 3A). If staple crossovers are installed every 21 bp along each inter-helix interface, then there are exactly four staple paths, of length 1848 bases, two traversing from one end to the other from 5' to 3', and two traversing in the opposite direction from 5' to 3'. We define the z -coordinate as the distance along each helix from the “front” (*i.e.* left-hand side of caDNAno file) to the “back” (*i.e.* right-hand side of caDNAno file). The staple paths of 24hb follow a very regular pattern, whereby they wind back and forth within a 7 bp z -window along five of the outer helices, and then wind back and forth within the next 7 bp z -window along seven helices through the interior of the helix bundle (Fig. S13–S17†). Therefore, an aesthetically pleasing pattern is produced by breaking each staple path into alternating

35mer and 49mer subpaths along these transitions between adjacent z -windows (Fig. 3B, version 1).

Surprisingly, when we implemented this staple-break pattern in 24hb, little or no productive folding could be observed, as indicated by gel electrophoresis as a lack of material migrating with the faster mobility associated with well folded structures (Fig. 3C, version 1). Similarly, if the staple-break pattern was altered so that each 35mer was lengthened by 3 nt on its 5' end and 4 nt on its 3' end compared to version 1, leading to a uniform length of 42 nt for almost every staple strand (Fig. 3B, version 2), still little or no productive folding could be observed (Fig. 3C, version 2). No significant improvement was observed for a longer (3 day) folding ramp. In contrast, when the staple-break pattern was shifted by 14 nt upstream compared to version 1 (equivalent to a shift 28 nt downstream due to the 42 nt pseudo-repeat) (Fig. 3B, version 3), folding then became enabled (Fig. 3C, version 3). Similarly, when the staple-break pattern was shifted by 24 nt upstream compared to version 1 (equivalent of 18 nt downstream) (Fig. 3B, version 4), then folding became enabled as well (Fig. 3C, version 4). The folding yields of version 3 and version 4 are estimated to be 22% and 25% respectively. The most obvious difference between the failed version 1 or version 2 designs and the successful version 3 or version 4 designs is that the latter two maintain a 14 nt uninterrupted stretch of complementarity to the scaffold, *i.e.* without any strand breaks or crossovers on either scaffold or staple segments. For version 0, each uninterrupted stretch is exactly 7 nt, while for version 1, each staple strand has a longest uninterrupted stretch of only 11 nt. This result is consistent with a model in which productive folding of 24hb requires staple strands to have a sufficiently long “seed” segment (*e.g.* 14 nt) that can nucleate specific attachment to the scaffold. If the staple strands lack these seeds, we hypothesize that folding fails because a family of kinetically trapped misfolded intermediates is the dominant product of the thermal-ramp folding procedure. Without further investigation, however, we cannot rule out the possibility that the origin of the misfolding is thermodynamic rather than kinetic in nature (see Discussion). We also observed the importance of 14 nt seeds for folding a more complex multi-layer origami structure, the honeycomb 8 × 8 64hb; these results are described in the Supplementary Information (Fig. S18–S23†). For all studies permuting the staple break points, the mapping of the scaffold sequence on to the design is not permuted.

In order to test the effect of folding 24hb where only a subset of the staple strands possess 14 nt seeds, we repeated the assembly with hybrid structures built from combinations of strands from the version 2 and version 3 designs. Each complete set of staple strands was divided into four subpools according to which of the four global staple paths it embodied. For each staple path of each hybrid structure, either the version 2 (*i.e.* non-seed-bearing staple strands that hinder folding) or the version 3 (*i.e.* seed-bearing staple strands that promote folding) subpool was selected. We tested all $2^4 = 16$ possible combinations, listed in Fig. 3D. Folding quality was assessed by relative agarose-gel mobility (Fig. 3E); as before, the fastest migrating bands with the mobility indicated by the red arrow represent the population of successfully folded structures.

For a 24 h thermal ramp, folding was rescued partially when 25% of the staple strands had a 14 nt seed, but rescued almost

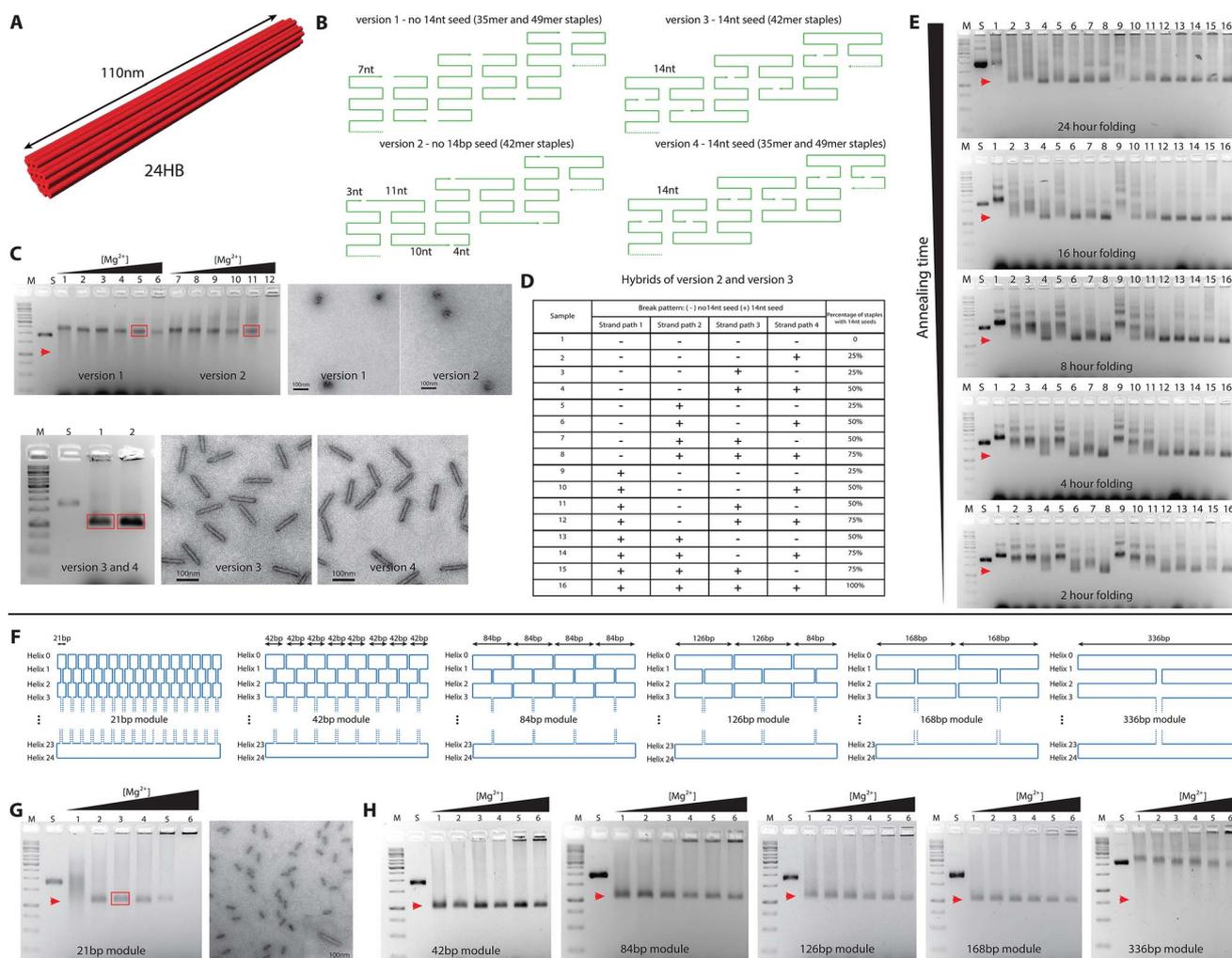


Fig. 3 Engineering kinetic access to productive folding of 24hb *via* staple-break selection: the importance of 14 nt seeds. (A) Schematic of 24hb DNA origami folded with p8064 scaffold. Each cylinder represents a double helix. (B) Different break patterns of staple strands. Four patterns of breaking points were applied to the four continuous staple-strand paths of the 24hb. Version 1 breaks each 14 nt segment into two 7 nt subsegments. Version 2 breaks each 14 nt segment into a 3 nt and an 11 nt subsegment, or a 4 nt and 10 nt subsegment as indicated. Versions 3 and 4 retain intact 14 nt “seeds” and only apply the breaks at select 7 nt segments. (C) Gel electrophoresis and TEM assays of 24hb with different staple-break designs. Top panel shows the unsuccessful folding by version 1 and version 2 staple-strand sets. Lane M shows 1 kb ladder. Lane S shows p8064 scaffold. Lanes 1 to 6 show the 24hb folded in the presence of 10, 12, 14, 16, 18, 20 mM $MgCl_2$. The red arrow indicates the position corresponding to the well folded 24hb particles. The bottom panel shows the well folded particles produced by annealing with version 3 and 4 staple strands. Lane M shows 1 kb ladder. Lane S shows p8064 scaffold. Lanes 1 to 2 respectively show version 2 and version 3 folded in the presence of 18 mM $MgCl_2$. Samples were extracted from the bands circled in the red boxes and subjected to TEM imaging. (D) Hybrid versions produced by mixing version 2 and 3 staple strands. Each hybrid version consists of 0, 25, 75 or 100% of staple strands with 14 nt seeds. This is achieved by mixing a subset of the staple strands lacking a 14 nt seed (version 2) with a subset of the staple strands with a 14 nt seed (version 3). Strand Path 1 to Strand Path 4 refer to the “continuous” staple-strand paths that exist before application of breaks. “-” indicates version 2 design with no 14 nt seed. “+” indicates version 3 design with one 14 nt seed for each staple strand. (E) Study of the folding of hybrid versions of 24hb as a function of annealing time (from top to bottom: 24, 16, 8, 4, or 2 h folding ramps). Versions with higher percentage of staple strands with 14 nt seeds show better folding quality, given the same folding time. Arrows indicate band mobility corresponding to the well-folded 24hb particles. (F) Designs of the scaffold path with increasingly larger modules. (G) Gel electrophoresis and TEM assays of 24hb with 21 bp modular scaffold design. Lane M shows 1 kb ladder. Lane S shows p8064 scaffold. Lanes 1 to 6 respectively show 24hb folded in the presence of 10, 12, 14, 16, 18, 20 mM $MgCl_2$. The arrow indicates band mobility corresponding to well folded 24hb particles. The 24hb sample was extracted from the bands circled in the red boxes and subjected to TEM imaging. (H) Gel electrophoresis assays of 24hb with 42 to 336 bp modular scaffold designs. Lane M shows 1 kb ladder. Lane S shows p8064 scaffold. Lanes 1 to 6 respectively show 24hb folded in the presence of 10, 12, 14, 16, 18, 20 mM $MgCl_2$. The arrows indicate band mobility corresponding to well folded 24hb particles. Scale bars for TEM images in (C) and (G) are 100 nm.

completely when 50% or 75% of the staple strands had a 14 nt seed. However, for more demanding folding conditions, simulated with compressed thermal ramps down to two hours, folding was rescued only partially when 50% of the staple strands had

a 14 nt seed, and the best folding was obtained when 100% of the strands had a 14 nt seed. Thus it appears the best folding can be achieved under more stringent conditions when as many staple strands as possible have a 14 nt seed.

Contact order for DNA origami refers to the average sequence distance between scaffold segments that are brought into close proximity by staple strand interaction; greater average distances correspond to higher contact order. For M13-scaffolded 24hb, we examined how contact order affects folding. We designed versions with the scaffold folded into modules of length 21 bp, 42 bp, 84 bp, 126 bp, 168 bp, and 336 bp (Fig. 3F, S24-S30†) (note that the structures described in Fig. 3B were routed with the scaffold in 168 bp modules). The only version that was not able to fold efficiently was the one with 336 bp modules. This observation is consistent with a model where a large contact order can create difficulties for multi-layer DNA origami folding. The threshold of the contact order where it becomes limiting for folding may vary from one architecture to another; for example, the Rothmund rectangular origami, a single-layer 2D design, consists of a single module of length 288 bp, and this design folds well; the threshold may be shorter in length for complex 3D multi-layer designs.

We further investigated the importance of seeds for folding 24hb in the context of a new design strategy for DNA origami: use of “short scaffold-parity” strands. In the original presentation of DNA origami, one long scaffold strand and many shorter staple strands are employed, where the only programmed complementarity occurs between scaffold- and staple-parity segments. The use of a long strand to scaffold small oligonucleotide-based tiles also has been reported.²⁰ Here we implemented a fraction of the scaffold-parity segments as short oligonucleotides that we dub “short scaffold-parity” strands.

We first tried three different implementations of short scaffold-parity strands to fold 24hb. For all versions, the scaffold used was the M13-based p7308. Each double helix contains 308 base pairs (and 28 total unpaired scaffold bases) distributed between two 168 bp scaffold modules. Each short scaffold-parity strand is 42 nt long and based on a randomly-generated sequence. Most staple strands are 42 nt long and contain a 14 nt seed. In addition, 1260 bases of the p7308 scaffold were left unstapled on one end of the 24hb (Fig. S31–S34†). In version 1 (Fig. 4A), the outer eighteen helices are populated by p7308, and the inner six by the short scaffold-parity strands. In version 2 (Fig. 4B), p7308 traverses a three-leaf clover path covering eighteen helices, leaving six outer helices to be populated by short scaffold-parity strands. In version 3 (Fig. 4C), p7308 traverses an accordion-style folding through twenty helices, leaving four outer helices for occupancy by short scaffold-parity strands.

All three versions produced successful folding, although the yields varied dramatically (Fig. 4D–4F). The major mode of folding failure was manifested by gel analysis as multimerization, and the least unwanted multimerization was seen for version 1. This improved folding coincides with the design of the short scaffold-parity strands as sequestered on the interior of the nanoparticle. The unwanted multimerization observed for designs where the short scaffold-parity strands are displayed on the exterior of the nanoparticles is consistent with unwanted short scaffold-parity strand exchange with other like particles in solution. This result suggests that the implementation of short scaffold-parity strands only will lead to high-yield assembly in cases where the unwanted strand exchange can be suppressed. One method for such suppression is to avoid designing short scaffold-parity strands on the exterior of the nanoparticles. However, other

methods for suppressing strand exchange for short scaffold-parity strands on the exterior surface may be possible; for example, particles perhaps could be sequestered inside of larger “folding compartments” that prevent them from encountering each other.

We implemented a fourth version (Fig. 4F) where the twelve outermost scaffold helices are populated by p7308 that jumps through unpaired scaffold segments that make parallel crossovers moving from one pair of outer scaffold helices to the next (Fig. S35–S40†). Then the six “inner” helices and six “middle” helices are populated by short scaffold-parity strands. When the staple strands are designed with no 14 nt seeds (Fig. 4G, version i), the structures fail to fold properly (Fig. 4H, lane 1; Fig. 4I, first TEM image). In contrast, successful folding can be obtained when the staple strands contain 14 nt seeds. However, significant unwanted multimerization still is observed. This result suggests that short scaffold-parity strands on the “middle” helices still can participate in unwanted strand exchange. Interestingly, greater unwanted multimerization is observed for staple strands where the 14 nt seed is placed near the middle of each staple strand (version ii) *versus* staple strands where the 14 nt seed is placed on the 5' end of each staple strand (version iii) (Fig. 4H, lane 2 *versus* lane 3). Placement of the seeds on the end of the staple strands may enable more rapid nucleation, which then may allow productive folding to occur more quickly compared to unwanted multimerization.

Discussion

We previously found that design of honeycomb 6×10 multi-layer 60hb origami folded to higher yield when the helices were designed to be underwound to 11 bp turn^{-1} , and speculated that the resulting underwinding strain was more than compensated by energetic relief from the ability of adjacent helices to bow further away from each other, due to the greater length between crossovers, and therefore the reduced electrostatic repulsion;⁵ an additional origin of folding improvement could have derived from the overall longer staple strand length distribution, or else the greater number of 8 bp anchorages *versus* 7 bp anchorages. However, for the 24hb under investigation for this study, the design underwound to 11 bp turn^{-1} did not fold as well as the design at $10.5 \text{ bp turn}^{-1}$. The different behavior between these 60hb and 24hb designs suggests that the net energetic benefit for introducing underwinding into DNA nanostructures is architecture dependent. For the 24hb, we observed that the best folding result was achieved with a 12 bp turn^{-1} design and annealing in the presence of $8 \mu\text{M}$ ethidium. Folding under these optimized conditions appeared to be mildly better than that achieved for the default $10.5 \text{ bp turn}^{-1}$ folding in the absence of an intercalator. For this 12 bp turn^{-1} design, the underwinding strain was too great to support folding in the absence of an intercalator. Conversely, the presence of the positively charged intercalator stabilized the underwound conformation and provided charge neutralization, and further increased the length between crossovers, which could have led to even greater relief of electrostatic repulsion due to the increased bowing of helices allowed. One prediction from this result is that some DNA-origami architectures, perhaps the 6×10 60hb origami discussed above, may exhibit markedly improved folding from a 12 bp turn^{-1} design folded in the presence of an intercalator, compared to a default $10.5 \text{ bp turn}^{-1}$ design.

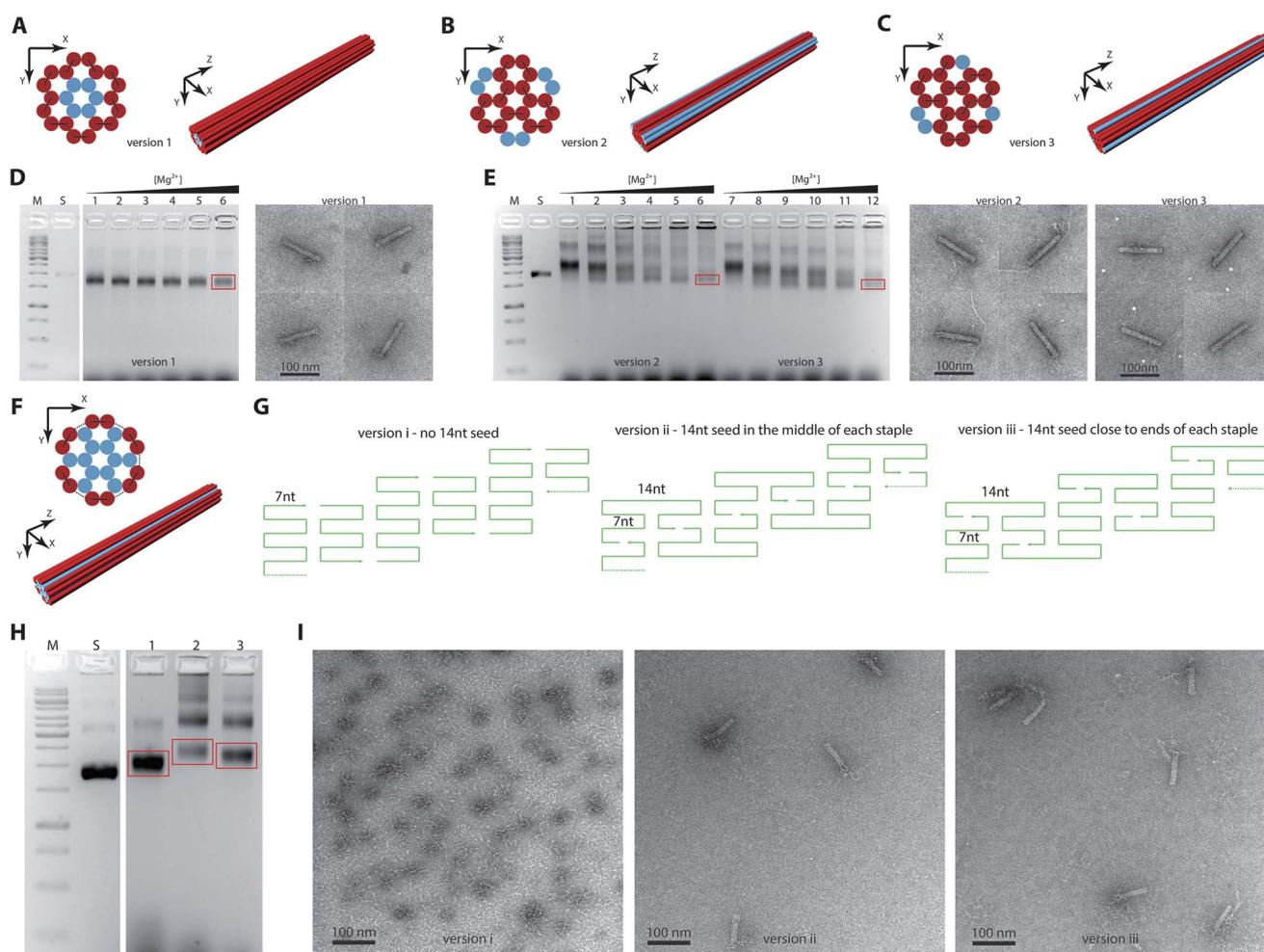


Fig. 4 Multi-layer DNA origami folding with a mixture of long and short scaffold strands. (A) to (C) Designs of 24hb with short “short scaffold-parity” strands. Cylinders in red indicate the DNA duplexes formed by p8064 scaffold and staple strands. Cylinders in blue indicate the DNA duplexes formed by short scaffold-parity strands and staple strands. (D) and (E) Gel electrophoresis and TEM images of version 1, version 2, and version 3 designs. Lane M shows 1 kb ladder. Lane S shows p8064 scaffold. Lanes 1 to 6 respectively show 24hb folded in the presence of 10, 12, 14, 16, 18, 20 mM $MgCl_2$. Well folded structures were extracted from the bands circled in the red boxes and subjected to TEM imaging. (F) A design of 24hb with twelve outside helices formed by p8064 scaffold and staple strands and twelve inside helices formed by short scaffold-parity strands and staple strands. (G) Three versions of staple-break designs for the 24hb shown in (F). Version i breaks each 14 nt segment into two 7 nt subsegments. Version ii and version iii both retain intact 14 nt seeds. The 14 nt seeds are close to the middle of each staple strand for version ii, while the 14 nt seeds are close to the 5' end of each staple strand for version 3. (H) Agarose gel electrophoresis of the three designs shown in (G). Lane M shows 1 kb ladder. Lane S shows p8064 scaffold. Lanes 1 to 3 respectively show the folding results of version i, version ii, and version iii. Structures were extracted from the bands circled in the red boxes and subjected to TEM imaging. (I) TEM images of the structures of version i, version ii, and version iii. Scale bars are 100 nm.

Origami designs with greater than 12 bp turn^{-1} reciprocal twist density could be partially but not fully rescued in their folding by increased concentrations of intercalator, due to a competing aggregation behavior. One possible contributing factor is that, for the 12 bp turn^{-1} design, a uniform incorporation of ethidium intercalators, one every six bp, is predicted to stabilize the structure, explained as follows: each ethidium intercalation has a preferred underwinding of the double helix of -26° (*cf.* $360^\circ / (10.5 \text{ bp turn}^{-1})$ or 34.3° equilibrium twist per base pair). So two intercalation events every 12 bp gives rise to an equilibrium twist of $12 \times 34.3 - 2 \times 26 = 360^\circ$. However, for higher reciprocal twist densities (*e.g.* $13.5 \text{ bp turn}^{-1}$), such a uniform preferred intercalation pattern does not exist; a non-uniform intercalation pattern may introduce more strain microscopically than a uniform intercalation pattern.

The ability to construct robust DNA nanostructures with a reciprocal twist density of 12 bp turn^{-1} , stabilized by intercalator binding, could serve as a useful capability. Here the twist per base pair is a round number: 30° . For example, the hexagonal close packed lattice structures²¹ perhaps could be implemented with $10 \text{ bp} = 300^\circ$, and therefore six 10 bp steps would enable crossovers to all six nearest neighbors with a repeating 60 bp , five-turn pattern. It also may be possible to suppress supertwisting in paranemic crossover structures^{22,23} that have 7 bp per major groove spacing and 5 bp per minor groove spacing.

We have found here that trisacridine ligands exhibit much stronger binding to underwound DNA nanostructures than to torsionally relaxed DNA nanostructures. This behavior is consistent with the known propensity of intercalators to unwind DNA

upon binding. We have exploited this specificity to develop a new approach to engineering biomimetic surfaces on underwound DNA nanotubes or else to the convex face of DNA nanotube rings, where the DNA is designed to be locally underwound. Attachment of inert guests to the trisacridine, such as PEG, provides a route to homogenous coating of the DNA nanostructure to provide passivation, which may prove useful for antifouling and improved biocompatibility. We anticipate that this method should prove readily adaptable to coat DNA nanostructures with a large variety of molecules such as polymers, proteins, and ligands.

Here we reported a second design strategy to improve the folding yield of a multi-layer DNA origami nanostructure: shifting of the staple-break distribution such that as many staple strands as possible include a 14 nt seed segment. As the number of staple strands was not changed by this shift, it seems more likely to us that there was little change in the thermodynamic stability of the structures, and that the improvement in folding was due almost entirely to kinetic factors; however, conclusive resolution of this issue awaits further analysis. If kinetic factors dominate, then a route to ordered assembly becomes evident. For example, the subset of staple strands bearing 14 nt seeds would be expected to bind to the scaffold earlier than strands without 14 nt seeds. Furthermore, altering the binding energy of the seeds, either *via* increasing the length or else changing the sequence, may enable finer control over order of assembly, *e.g.* direct folding of outside before inside, or *vice versa*. The exact length requirements for the seeds likely will be both sequence and architecture dependent; this should prove an interesting topic of investigation for a later study. The observation that the placement of seeds on the end of strands appears consistent with the increased binding kinetics observed when toeholds are placed on the ends of strands. The presumed mechanism is that swivelling of double helices where one strand is short is much faster due to frictional considerations.²⁴

Finally, we have found that the scaffold fraction of DNA origami can be distributed over a single long strand and multiple shorter “short scaffold-parity” strands. This capability should add flexibility in DNA origami design, especially the ability to construct shapes that are much larger than is allowed if all the scaffold material is required to be on a single strand, as the production of such long scaffold strands presents additional technical challenges. Furthermore, longer scaffold strands may be subject to greater tangling behavior. However, we have observed that placement of short scaffold-parity strands on the exterior of 3D DNA origami can lead to greater unwanted aggregation, perhaps due to a strand exchange mechanism. Therefore short scaffold-parity strands are better suited to designs where such strand exchange is discouraged, such as on the interior of the object, or else in conjunction with some other folding environment where unwanted multimerization is otherwise inhibited.

Methods

Synthesis and characterization of PEG-tris-acridine conjugates

Coupling of PEG20k-NHS and dendrimer. 5.0 μmol of PEG20k-*N*-hydroxysuccinimide (NHS) was dissolved in 2 mL dichloromethane and slowly added to a stirred 100 μmol 25% aqueous solution of G1.0 Polypropylenimine tetramine with 10% dichloromethane (DCM). After 30 min of stirring at 20 °C, after

60 min, and after 90 min, 42 μmol dry EDC•HCl was added to the reaction mixture. After 2 h total reaction time, solvent was removed *in vacuo*. The product was purified by gel-filtration on a Superdex 75 column (GE Life Sciences), concentrated *in vacuo* and immediately used in the next step.

Acridine-isothiocyanate synthesis. 9-chloroacridine (1 mmol, 214 mg) and AgSCN (2 mmol, 332 mg) was dispersed in 3 mL toluene in a closed 4 mL glass vial with thorough stirring and incubated in a 110 °C oil bath for 24 h. After 24 h, the stirring of the reaction mixture was stopped and the solids were allowed to drop to the bottom of the vial. The clear colorless liquid was transferred to a glass vial and DCM was added to the reaction vial to extract more product from the solids. The DCM washing was repeated 3–4 times.

PEG-dendrimer coupling to acridine-isothiocyanate. The product from the previous step was re-dissolved in 2 mL methanol containing 70 μmol of triethylamine and 1 mL of the acridine-isothiocyanate product solution was added. After mixing, the solution was incubated at 20 °C for 30 min. 1 mL of acridine-isothiocyanate was added two more times along with 0.1 mmol of triethylamine and the mixture was left to react for 30 min each cycle. After reaction, water and *n*-BuOH were added to have the reaction mixture separate into an organic and an aqueous phase, with the organic phase holding the excess acridine and the aqueous phase the PEG-conjugate. The aqueous phase was extracted several times with BuOH, until the BuOH-phase was not yellow anymore. After extraction, the yellow aqueous phases were lyophilized. Yellow PEG-dendrimer-acridine conjugate was purified by gel filtration using a Superdex 75 column in water and by reverse phase HPLC C18 column (Sigma Aldrich). UV absorbance of these conjugates at 420 nm was measured to calculate the acridine content. MALDI-TOF mass spectrometry was used to determine an average MW of these conjugates.

Design and assembly of DNA origami structures

The DNA origami nanostructures were designed using the honeycomb-lattice version of the caDNAno software (<http://www.cadnano.org/>). Assembly of the DNA nanostructures was accomplished in a one-pot reaction by mixing scaffold strands derived from M13 bacteriophage at 10 nM with 50 nM each oligonucleotide staple strand (reverse-phase cartridge purified, Bioneer Inc.) in a folding buffer containing 5 mM Tris, 1 mM EDTA, pH 8, and a concentration of MgCl₂ noted below, and subjecting the mixture to a thermal-annealing ramp: 80 °C for 5 min, –1 °C per cycle for 14 additional cycles, 65 °C for 20 min, –1 °C per cycle for 35 additional cycles. MgCl₂ conditions listed in parenthesis: 6hb_10.5 bp turn⁻¹ (20 mM), 6hb_ring (20 mM), 12hb_10.5 bp turn⁻¹ (16 mM), 12hb_11.0 bp turn⁻¹ (16 mM). For other structures, the MgCl₂ was varied as listed in the figures, and structures were formed by quickly cooling the oligo mixtures from 80 °C to 60 °C over 1 h, then continue cooling from 60 °C to 24 °C over different time course of 2 h, 4 h, 8 h, 16 h, 24 h or 72 h. The folding ramps were named according to the time of the second cooling step, as 2-hour ramp, 4-hour ramp, *etc.*

Agarose gel analysis

Annealed DNA structures were purified by gel (1.5% to 2% agarose, 0.5xTBE (45 mM tris-borate, 1 mM EDTA, pH 8.3), 11 mM MgCl₂, 0.5 μg mL⁻¹ ethidium bromide) electrophoresis with a Thermo Scientific Owl B2 EasyCast Mini Gel System apparatus in an ice-water bath. The DNA structure samples were loaded into the agarose gel and allowed to migrate for 3 h (running buffer: 0.5x TBE, 11 mM MgCl₂; running voltage: 2.85 V cm⁻¹). The fast migrating bands corresponding to the correctly folded structures were then visualized with ultraviolet light and cut out from the gel. Excised bands were crushed and transferred into DNA gel extraction spin column (BIO-RAD, Catalog number: 732-6166). The DNA structure solution was recovered by centrifugation of the loaded column for 10 min at 16000 × g. Yields of DNA nanostructures are estimated from fluorescence intensity. First the intensity of target band and that of the entire lane were measured using software ImageJ. After background correction, the yield is calculated as the ratio between the two, *i.e.* (Yield = Intensity of nanostructure band/Intensity of entire lane).

Coating and agarose gel mobility shift assay

The intercalation complex were prepared by mixing DNA with synthesized HPLC-purified PEG-tris-acridine in 10mM Tris, 1 mM EDTA, pH 8.0, 20 mM MgCl₂, at the weight ratios indicated in the text, and incubated at 25 °C for 30 min.

Incubated particles were electrophoresed on a 1.5% agarose gel containing 0.5x TBE buffer (45 mM Tris boric acid, 45 mM Tris base, 11 mM MgCl₂ and 1 mM EDTA (pH8)) at 2.5 V cm⁻¹ for 3 h. After electrophoresis, the gel was stained by 0.5 μg mL⁻¹ of ethidium bromide and analyzed by a Typhoon biomolecular imager (Typhoon Trio™ Variable mode imager, GE Healthcare) following the fluorescence of ethidium bromide at 610 nm band-pass 30 nm width emission filter and acridine at 520 nm band-pass 40 nm width emission filter.

Transmission electron microscopy

For imaging, particles were adsorbed onto glow discharged collodion and carbon coated TEM grids (Ted Pella) and then stained using a 2% aqueous uranyl formate solution containing

25 mM NaOH and visualized at 68000× magnification with an FEI Tecnai T12 BioTWIN or a JEOL JEM-1400 transmission electron microscope, operated at 80 kV in the bright field mode. For PEGylated nanostructures, 0.5% ruthenium tetroxide aqueous was used to stain synthetic polymers like PEG.

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